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Composition of Ge(Si) islands in the growth of Ge on Si(111)

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X-ray photoemission electron microscopy (XPEEM) is used to investigate the chemical composition of Ge/Si individual islands obtained by depositing Ge on Si(111) substrates in the temperature range 460–560 °C. We are able to correlate specific island shapes with a definite chemical contrast in XPEEM images, at each given temperature. In particular, strained triangular islands exhibit a Si surface content of 5%–20%, whereas it grows up to 30%–40% for “atoll-like” structures. The island’s stage of evolution is shown to be correlated with its surface composition. Finally, by plotting intensity contour maps, we find that island centers are rich in Si. © 2004 American Institute of Physics. [DOI: 10.1063/1.1758304]

Current attempts to advance the existing Si technology and to integrate micro- and optoelectronic devices on the same Si wafer have spurred significant efforts, directed toward the possibility of growing Ge nanostructures on silicon, with particular interest in quantum dots (QDs).1–4 The aim is to exploit the electronic properties of these novel systems to engineer completely new components that are compatible with today’s Si-based devices. In this context, the growth of QDs by self-assembly is an emerging technology.

There are at least three critical issues to be addressed in the growth of self-assembled semiconductor QDs, namely: (i) the controlled positioning of QDs on a suitable substrate;5,6 (ii) the ability to grow stable dots with the uniform size and shape;7 (iii) the control of the composition of individual QDs, which ultimately determines their physical properties.8–11

Intermixing has been previously investigated in different systems, either statistically from sets of islands,12 or by analyzing the cross section of single QDs.13 In this letter, we report spectroscopy-beamline measurements of Ge/Si alloying from the surface of individual Ge islands grown on Si(111), which allows the representation of their stoichiometry by means of two-dimensional contour maps. The islands nucleate as a result of a modified Straniski–Krastanov growth process,14 when Ge coverage exceeds ~3 monolayers (ML).

After degassing at 600 °C for several hours, Si(111) was flashed at ~1200 °C in ultrahigh vacuum conditions. Three samples were grown at substrate temperatures of 460, 530, and 560 °C, depositing ~10 ML of germanium by molecular beam epitaxy. The morphology and chemical contrast of the grown structures was studied at room temperature by x-ray photoemission electron microscopy (XPEEM) at the Nano-

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the first phenomenon is predominant, then a Si-rich island core should result, as a consequence of atomic exchange processes at its base during growth. Conversely, if surface diffusion processes are dominant then alloying should prevail at island edges. Experimental studies on different systems at different growth conditions have alternatively shown a Si-rich center or perimeter. In the present case, which confirms bulk driven alloying, we can interpret the Si-rich central zones as a step in the pathway toward island evolution, until atolls appear.

When photoelectron yields from very close regions of the same image are directly compared, and the spectra from all atomic species present on the sample are available, the chemical composition of the surface can be evaluated. If we name the ratios between the Si 2\textit{p} and the Ge 3\textit{d} yields from the islands and from the WL in proximity of the islands as \( R_{\text{Si}} \) and \( R_{\text{Ge}} \), respectively, and relate them through the island surface concentration, the Si concentration \( x_{\text{Si}} \) may be expressed as a function of \( R_{\text{Si}} \) and \( R_{\text{Ge}} \) through the simple relation:

\[
    x_{\text{Si}} = \frac{R_{\text{Ge}} - 1}{R_{\text{Ge}}/R_{\text{Si}} - 1}.
\]

Equation (1) allows one to estimate the islands’ Si surface concentrations \( x_{\text{Si}} \) from a set of individual islands grown at a given substrate temperature and to relate them to their geometrical features. Figure 3 displays a graph of Si concentration plotted as a function of island base area. The plot clearly shows that islands with different morphological and chemical characteristics coexist at the three different growth temperatures. The following description is thus suggested:

**FIG. 1.** 4×4 μm\(^2\) integrated XPEEM images taken in correspondence of (a) the Si 2\textit{p} core level peak and (b) the Ge 3\textit{d} core level. Examples of spectra are shown in the insets. Circles: raw data; shaded curves: background subtracted data, averaged over the labeled regions. The micrographs are obtained by integrating the spectra with ~20 nm lateral resolution. The x-ray energy was 130.5 eV. The growth temperature was 560 °C. The measured Si concentrations for the 3D islands are labeled in panel (a).

**FIG. 2.** (Color) 2×2 μm\(^2\) integrated XPEEM image taken at the Si 2\textit{p} core level, together with the contour plots from a more (top) and a less (bottom) ripened island. Photoelectron yields are increasing from blue (lowest) to red (highest). The darkest regions in the panels are produced by the shadows of the 3D islands, due to the 16° grazing incidence angle of the x-ray beam. The growth temperature was 530 °C. The estimated surface Si concentrations for the selected islands are indicated. The wetting layer appears to be inhomogeneous.

**FIG. 3.** (Color) Base area dependence of the Si surface content in the observed islands. At each deposition temperature, the stoichiometry is determined by the island’s lateral dimensions.
the ripening stage of an individual island can be equivalently associated with its geometrical features or to the island’s Si surface concentration, at a given growth temperature. The latter defines the details of the stoichiometry–morphology relation in the single 3D islands and determines the population of the coexisting species. Most islands have a Si concentration ranging from 5% to 20% and a reduced area or high aspect ratio, whereas ripened islands are typically characterized by a value of $x_{Si}$ greater than 25%. The measured Si concentrations are consistently lower than the 50% value previously reported for the WL. Increasing the temperature alters the relative population of the two species, privileging ripened islands, also shifting the surface Si concentration of strained islands to higher values. We remark that the data sets obtained at 460 and 530°C do not exhibit major changes, while evident alterations of $x_{Si}$ values are found at 560°C. This is clearly related to the kinetic activation of the diffusion processes between the substrate, the WL, and the 3D islands. However, this does not allow us to establish whether they occur predominantly at the surface or at the underlying layers (e.g., through vacancy migration). Moreover, the outlining of two distinct island species at these temperatures points to the existence of two metastable configurations, which rules out an island evolution caused by Ostwald ripening alone. Nevertheless, it was previously shown that room temperature imaging does not allow one to relate the observed statistics to island thermodynamics at a given deposition temperature.

In conclusion, we have shown that specific geometrical features are unambiguously related to a definite chemical contrast in XPEEM images in the temperature range 460–560°C, and thus to a specific Si/Ge composition at the island’s surface. XPEEM yields quantitative information and leads to surface concentration–morphology graphs for sets of individual islands. This technique can also be used to plot contour maps which indicate concentration gradients at the surface of single Ge(Si) islands. The resulting values point to the presence of two island configurations: strained triangular islands characterized by a Si surface concentration of 5%–20%, and ripened atolls that contain up to 30%–40% Si. Higher growth temperatures induce faster ripening of the 3D islands and a shift of the surface Si content in the strained islands toward higher values, indicating a thermally activated interdiffusion process that partially relaxes the strain energy in the islands. This set of observations may be used to control the composition/morphology of self-assembled semiconductor QDs.

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8. It was shown that intermixing occurs since the very first stages of Ge deposition on Si. See, e.g., P. Castrucci, R. Gunnera, M. De Crescenzi, M. Sacchi, G. Dufour, and F. Rochet, Phys. Rev. B 60, 5759 (1999).
17. The sensitivity of the technique in the direction perpendicular to the surface plane depends on the escape length of the excited electrons at a given kinetic energy. The escape length for the Si 2p and Ge 3d photoelectrons at the selected x-ray wavelength is ~0.5 nm. Thus XPEEM images show the composition of the uppermost surface layers of the sample.
19. The darkest areas in the images represent an artifact due to the geometrical setup of the experimental system: they are related to the shadowing effect of the incident x rays caused by the 3D islands themselves. In principle, the shadow’s length allows one to estimate the island height.
26. The high surface sensitivity of XPEEM does not allow one to describe the alloying dynamics in the buried layers.
27. We start by integrating the formulas for the photoelectron yields from an infinitesimal volume element of the sample (Ref. 20) over the coordinate perpendicular to the surface. The Si concentration profile within the island, WL, and substrate was modeled as a constant function with values $x_{Si, WL}$, and $x_{Si, substrate}$, respectively.
29. Due to the low value of the photoelectrons’ escape length, the measured Si concentrations $x_{Si}$ can be interpreted as an average value, with exponential weights, over the surface of individual 3D islands.